#### **Nanometer-Scale Four-Point Probe Resistance Measurements of Individual** 2 Nanowires by Four-Tip STM 3

S. Hasegawa, T. Hirahara, Y. Kitaoka, S. Yoshimoto, T. Tono 4 5 and T. Ohba

Abstract We present a review of our recent results about transport properties of 6 nanowires measured by a four-tip scanning tunneling microscope (STM) installed 7 with metal-coated carbon nanotube (CNT) tips. We first present our custom-made 8 apparatus (with UNISOKU Co.) as well as CNT tips, and then some case studies 9 with two different samples, Co-silicide nanowires self-assembled on Si(110) sur-10 face and Cu nanowires made by damascene processes used in LSI industry. It is 11 shown that the four-tip STM with CNT tips is versatile and powerful for measuring 12 the conductivity of individual nanostructures. 13 14

#### **1** Introduction 15

Conductivity measurements in sub-micron or nanometer scale are of great interest 16 in nanoscience and nanotechnology. For example, nanoelectronics such as semi-17 conductor devices requires low and stable electrical resistance of interconnects to 18 maintain device performance. Several kinds of methods to measure the conduc-19 tivity at nanoscales have been developed including fixed electrodes made by 20 microlithography techniques. A method which adopts tips of scanning tunneling 21

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Fig. 1 Schematic drawings of the whole system of low-temperature four-tip STM (a) and around the STM stage (b) [14]. c A close-up photo of the STM stage without radiation shields. d The photo of the whole system

microscope (STM) as electrodes, however, has great advantages in positioning of 22 the probes in arbitrary configurations as well as in high spatial resolution of 23 measurements. But single-tip STM is not enough for versatile measurements 24 of transport properties at various kinds of nanostructures: we need source, drain, 25 and gate electrodes. For this reason, several groups [1-10] including companies 26 [11] and our group [12–14] have developed four-tip STM. In which four inde-27 pendent STM tips are operated in an organic manner with aid of a SEM or optical 28 one, and they are used as electrodes for microscopic two- or four-point probe 29  $(\mu 4PP)$  conductance measurements. In this article, we show our apparatuses 30 including installation of carbon nanotube tips [15-17] and some results of resis-31 tance measurements of nanowires obtained in our group. 32

## 33 2 Four-Tip STM System

Figures 1a, b show schematic drawings of our new version of four-tip STM system 34 [14], consisting of a main (STM) chamber, a sample preparation molecular beam 35 epitaxy (MBE) chamber, and two load-lock chambers for sample and tip 36 exchanges, all of which are UHV compatible. The STM tips can be installed into 37 the main chamber from the tip load-lock chamber where a hot W filament is 38 installed for out-gassing of the tips. The sample is introduced from the MBE 39 chamber where cleaning of the sample, deposition of materials and reflection-high-40 energy electron diffraction (RHEED) observation can be done. The sample can be 41 heated by direct current heating and cooled down to about 30 K by continuous-42 flow type cryostat in the MBE chamber. These capabilities are necessary for 43

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preparing aimed surface superstructures, epitaxial thin films, nanodots, nanowires, and for making in situ measurements. 45

The STM stage is mounted on the thermal conducting Cu rod which is soaked in the coolant of the bath cryostat below. The STM stage including the sample and four sets of actuator units is wholly surrounded with two-fold radiation shields and movable shutters. The photo of Fig. 1c is the stage without the radiation shields, and Fig. 1d shows the whole system. The sample and tips can be cooled down to 7 K and can be kept for 23 h with liquid He as coolant. In the case of liquid  $N_2$ , the minimum temperature is 80 K and the preserved time is longer than three days.

The SEM column (APCO Mini-EOC) is mounted above the STM stage. The 53 working distance of SEM is about 25 mm. The electron beam is irradiated from 54 SEM column through a 1 mm diameter hole in the radiation shields. A multi-55 channel plate for the secondary electron detection for SEM imaging is placed on 56 the inside wall the outer shield. The SEM image is obtained from the secondary 57 electron signal or beam induced current signal electron-beam-induced current 58 image (EBIC). The resolution of the SEM is about 20 nm for both signals. 59

A spring vibration isolator and an eddy-current damper are built between the 60 thermal conductor and the STM stage to avoid vibration of STM stage. The spring 61 isolator decoupled the STM stage from other components. However, during SEM 62 observation, tip/sample exchanges, and cooling the stage, the STM stage is fixed to 63 the thermal conductor (Cu rod) and therefore the isolator and damper are disabled. 64 When we fix the STM stage, the Cu plate works as a thermal conductor and 65 enlarges the contact area for good thermal connection. When we float the STM 66 stage by the springs, this plate makes eddy-current damper. Since alternative 67 arrangements of the small magnets make closed magnetic paths, the magnetization 68 does not affect the SEM beam. 69

Four sets of tip actuator units are mounted at the corner of the square STM 70 stage, and a sample actuator unit is placed at the center. The actuator units consist 71 of stacked piezo ceramics supported by sapphire plates. For fine positioning or 72 scanning in nanometer or sub-nanometer range, tips and samples are driven by 73 conventional piezoelectric effect of the ceramics by DC voltage. The maximum 74 positioning range by this method is about 2 µm to each direction. For coarse 75 positioning, the actuators are driven by stick-slip mechanism in 5 mm travel 76 distance in XY directions and 2.5 mm in Z direction at accuracy of about 100 nm. 77 In addition to these three- or two-dimensional-motion actuators, the tip actuators 78 also contain small piezo ceramics near the tips for fast STM feedback. 79

Figure 2 shows a series of SEM images of the four tips arranged in various 80 configurations [12, 13, 18]. The tips are chemically etched W wires. The probe 81 spacing can be changed from 1 mm to ca. 200 nm in Figs. 2a-c. They can be 82 arranged on a line equidistantly (linear  $\mu$ 4PP method) (c, d) in arbitrary directions, 83 or in square arrangement (square- $\mu$ 4PP method) Figs. 2e-h. The square can be 84 rotated with respect to the sample surface (rotational square- $\mu$ 4PP method) by re-85 positioning each tip under computer control. This is useful to measure anisotropic 86 surface conductivity in which the conductivity is different depending on the crystal 87 orientation. 88

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Fig. 2 SEM images of the four W tips in various arrangements in the four-tip STM [12, 13, 18]

# 89 **3 Metal-Coated Carbon Nanotube Tips**

An important issue for the four-tip STM is the probe spacing; the probe spacing 90 should be in the order of 10 nm to measure various kinds of nanostructures. At the 91 present the minimum probe spacing in the multi-tip STM is approximately 100 nm 92 when W tips are used. This is due to the radius of tip apex of electrochemically 93 etched W tips. This probe spacing is not small enough for observing ballistic 94 transport and quantum interference effects because the coherence length of con-95 duction carriers is shorter than 100 nm in many cases. For this reason, continuous 96 efforts are made to shorten the probe spacing down to ca. 10 nm. To make the 97 probe spacing shorter, carbon nanotube tips have been developed in which a 98 carbon nanotube is glued at the end of W tip [19-22]. Since the radius of the 99 (multi-walled) carbon nanotubes is usually ca. 10 nm and the aspect ratio is much 100 higher than usual W tips, two carbon nanotube tips can be brought together into 101 approximately 10 nm spacing. Another feature of the carbon nanotube tip is its 102 mechanical flexibility which can reduce damage to delicate samples such as 103 organic and biological molecules, and make the tips withstand numerous direct 104 contacts to the samples. These properties are quite convenient for the transport 105 measurements by multi-tip STM at nanometer scales. However, there have been 106 problems in the carbon nanotube tips; high contact resistance between the sup-107 porting metal tip and the attached carbon nanotube strongly disturbs electron 108 transport at the STM/STS measurements. Moreover, adsorbates contained in the 109 carbon nanotube degrade the surface cleanness of the specimen under STM 110 operation. 111

A novel technique for overcoming these difficulties has been developed; the carbon nanotube together with the supporting metal tip is wholly coated with a thin metal layer [15]. Figures 3a,b show TEM images of a W-coated carbon nanotube tip glued on a W supporting tip. The W layer of ca. 3 nm thick was deposited by pulsed laser deposition (PLD) method. The W layer fully covers the tip even at the

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Fig. 3 a TEM image and b high-resolution TEM image of a W-coated carbon nanotube tip coated by pulsed laser deposition method. c SEM images of the W-coated carbon nanotube tip, directly contacting to a sample surface. Reproduced from Ref. [15]. d A SEM image showing four carbon nanotube tips contacting a Co-silicide nanowire grown on a Si substrate [23]

end. Figure 3c shows flexibility and robustness of the W-coated carbon nanotube 117 tip upon the direct contact to a sample surface. Figure 3d is a SEM image showing 118 four CNT tips contacting a Co-silicide nanowire grown on a Si substrate [23]. We 119 have also confirmed that the electrical resistance at the glued point between the 120 carbon nanotube and supporting W tip is stably reduced by the metal coating; 121 especially PtIr coating is the most efficient for this purpose [16]. Atomic-resolution 122 STM imaging and STS spectra were acquired with the W-coated carbon nanotube 123 tip at the first attempt [15]. With this metal-coated carbon nanotube tips, we have 124 succeeded in bringing the two tips together into less than 30 nm [17, 23, 24]. Since 125 the resolution of SEM is not enough for observing a smaller probe spacing, we 126 believe that the minimum spacing can be reduced to ca. 20 nm, similar to the 127 diameter of CNT itself. 128

# 129 4 Measuring Co-Silicide Nanowires

CoSi<sub>2</sub> nanowires are known to grow self-assembly by depositing high-purity cobalt on a Si(110) clean surface held at 750–850°C in UHV, as shown in Figs, 4a–e [25]. The nanowires become longer and thinner with lowering the substrate temperature during the Co deposition. The wires are single-crystalline, half of which is embedded in the Si substrate as observed by a cross-sectional TEM image of Fig. 4d [25]. The CoSi<sub>2</sub> is known to be highly conductive metallic





**Fig. 4** a-c Atomic force microscopy images of  $\text{CoSi}_2$  nanowires formed by depositing Co on the heated Si(110) substrate at different temperatures reproduced [25]. **d** A cross-sectional TEM image of the nanowire from Ref. [25]. **e**-**h** SEM images of the inner pair of CNT tips (tip 2 and tip 3 in (j)), with different probe spacings, contacting one of the nanowires during the resistance measurements in the four-tip STM [23]. The current probes (tip 1 and tip 4) are about 1  $\mu$ m away from these voltage probes. **e**'-**h**' Four-terminal current–voltage curves measured at the tip configuration shown in **e**-**h**, respectively [23]. The four-terminal resistance  $R_4t$  decreased with reducing the probe spacing. **i** The voltage probes contacted each other, and the tip 3 was bent. **j** A schematic showing the four-terminal\_current–voltage measurements

and its resistivity is  $31 \pm 9 \ \mu\Omega$  cm for the nanowire [26] and ~15  $\ \mu\Omega$  cm for the films [27] at 300 K.

As shown in Fig. 3d, the four CNT tips were made contact onto one of the 138 nanowires under SEM observation. The tips were made approach beyond the point 139 of tunneling until the contact resistances became less than 1 M $\Omega$ , corresponding to 140 direct contact. At the current-voltage (I-V) measurement, the STM feedback loops 141 were cut. Even if the tip physically contacted the NW, the contact resistance 142 between the tip and sample was higher than 50 k $\Omega$ . It was difficult to reduce this 143 resistance because of the nanometer-sized contact area. This is much larger than 144 the resistance of the nanowire itself, which is less than 1 k $\Omega$  with probe spacing 145 smaller than 1 µm [26]. Therefore, by two-terminal I-V measurements, the 146 resistance did not depend on the probe spacing due to the large contact resistance 147 at the probe contacts: four-point measurements are indispensable at nanometer 148 scale. 149

Four-terminal I–V measurements were done by sweeping the bias voltage between tip 1 and tip 4 with recording the current I and the voltage drop V between tip 2 and tip 3, with changing the spacing between tip 2 and tip 3 as shown in Figs. 4e–h.

153 The SEM beam was stopped at the I-V measurements to avoid possible influence on

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Fig. 5 The measured fourterminal resistance  $R_4t$  as a function of the spacing between the voltage probes, and SEM image of the nanowire under measurement (*top view*). The *black arrow* around 600 nm in the graph and the *white arrow* in the SEM image indicate the position where the nanowire width changes, resulting in a change of the resistivity [23]



the resistance caused by high-energy electron beam (10 kV). Figures 4e-h show a 154 series of SEM images around the voltage probes (tip 2 and tip 3) touching on the 155 nanowire, and corresponding four-terminal I–V curves are shown in (e'-h'). 156 We reduced the probe spacing between the voltage probes during taking the I-V 157 characteristics. The positions of the two current probes (tip 1 and tip 4) and one of the 158 voltage probes (tip 2) were fixed in the measurements, and only tip 3 was shifted. All 159 I-V curves were linear. The four-terminal resistance  $R_4 t = dV/dI$  around I = 0 160 decreased with shortening the probe spacing. They are several  $\Omega$ , much smaller than 161 the contact resistance. A voltage amplifier was introduced at the STM pre-amplifiers 162 to detect small voltage drops resulted from the small resistance. Finally tip 3 bent as 163 shown in Fig. 4i, and  $R_4 t$  became 0  $\Omega$  because of direct contact between the voltage 164 probes. The minimum probe spacing achieved here was  $30 \pm 20$  nm as shown in 165 Fig. 4h. This was limited by the diameter of the CNT tip apex we used, 30 nm (20 nm 166 diameter of CNT + 5 nm thick PtIr layer). The error bar in the probe spacing is 167 determined by the radii of the apexes in tip 2 and tip 3. 168

We plot the measured four-terminal resistance  $R_4 t$  as a function of the spacing between the contact points of the voltage probes on the nanowire in Fig. 5. The linear proportional relation in the range 30–600 nm means diffusive transport, and the fit line gives one-dimensional resistivity  $\rho_{1D} = 57 \pm 3 \Omega/\mu m$ . By extrapolating the data points, there seems to be no residual resistance at zero probe

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Table 1	Probe	spacing l	L depende	nce of	the	four-terminal	resistance	$R_4t$ in	various	conduction
mechanis	sms									

Conduction mechanism1D Ohmic2D Ohmic3D Ohmic1D strong localization1D weak localizationBallisticL-dependence of $R_4 t$ $\propto L^1$ (constant) $\propto L^{-1}$ $\propto exp(\frac{L}{L_0})$ $\propto \frac{L}{L_0-L}$ $\propto L^0$ (with fluctuation							
<i>L</i> -dependence $\propto L^1 \propto L^0 \propto L^{-1} \propto \exp\left(\frac{L}{L_0}\right) \propto \frac{L}{L_0 - L} \propto L^0$ (with fluctuation	Conduction mechanism	1D Ohmic	2D Ohmic	3D Ohmic	1D strong localization	1D weak localization	Ballistic
	<i>L</i> -dependence of $R_4 t$	$\propto L^1$	$\propto L^0$ (constant)	$\propto L^{-1}$	$\propto \exp\left(\frac{L}{L_0}\right)$	$\propto \frac{L}{L_0 - L}$	

 $L_0$  is the localization length

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spacing, which is owing to the four-point probe configuration. The gradient decreased to  $19 \pm 4 \Omega/\mu m$  above 600 nm probe spacing. This is due to an increase of the nanowire width from  $100 \pm 20$  to  $160 \pm 20$  nm as shown in the SEM image in Fig. 5. By checking the reproducibility we found that the physical contacts of the CNT tips did not cause any significant damage to the nanowire.

We now discuss the transport property of the NW. Table 1 shows a list of 179 the probe spacing L dependence of the four-terminal resistance in various 180 conduction mechanism. The probe spacing dependence of resistance in CoSi<sub>2</sub> 181 nanowire showed a linear one-dimensional Ohmic feature ( $R_4 t \propto L$ ). This 182 behavior is due to a one-dimensional conduction path through the nanowire 183 without leakage of current to the underlying three-dimensional substrate or to 184 the two-dimensional substrate surface. This is because a Schottky barrier 185 between the nanowire and the Si substrate confines the current [26]. The mean 186 free path of the electrons in  $CoSi_2$  is around 6 nm at room temperature [28], 187 which is much smaller than the width and height of our nanowire as well as 188 the probe spacing. Therefore, our result of diffusive conduction is reasonable. 189 The three-dimensional resistivity of the nanowire can be calculated. The width 190 of the nanowire is determined by SEM image, and the height can be deter-191 mined by the transmission electron microscope image [25]. We obtain the 192 three-dimensional resistivity  $22 \pm 4 \mu \Omega$  cm. In the same way, we obtain 193  $19 \pm 4 \ \mu\Omega$  cm for the region larger than 600 nm. These values are comparable 194 to the previous results  $(31 \pm 9 \,\mu\Omega \,\text{cm})$  in which similar CoSi<sub>2</sub> NWs were 195 measured with W tips in larger probe spacing range [26]. 196

In the ballistic transport regime, two-terminal and four-terminal resistances 197  $(R_2t \text{ and } R_4t)$  do not depend on the probe spacing. They depend only on the total 198 transmission probability T23 of electron wavefunction between the voltage probes, 199 tip 2 and tip 3 (which are also the current probes in the two-terminal measure-200 ment). A remarkable feature of the ballistic transport is that  $R_4 t$  takes any value 201 between  $-R_2t$  and  $+R_2t$ , meaning that  $R_4t$  can be negative by quantum interference 202 effects [21]. At liquid-He temperature, the mean free path of conduction electrons 203 in a CoSi<sub>2</sub> film with the thickness of 110 nm becomes ca. 100 nm [27]. Therefore, 204 at low temperatures, we can possibly observe quantum interference effects in 205 resistance at probe spacing we achieved here by using the PtIr-coated CNT tips. 206 The probe spacing dependence of  $R_4t$  in the present experiment also excludes 207 observable effects of carrier localization. 208

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Fig. 6 Cross-sectional SEM image of damascene trenches. The trench widths are **a** 70 nm **b** 100 nm, and **c** 500 nm, respectively [29]

#### 209 **5 Measuring Cu Nanowires**

Copper (Cu) wires are now widely used for interconnects in semiconductor 210 devises. But, as the wire width reaches down to the sub-micrometer scale, which is 211 comparable to the mean free path of conduction electrons in the wires, a significant 212 increase in the resistivity has been observed. This is speculated as due to the 213 increased surface and grain boundary scatterings. As the wire width scales down, 214 electrons will undergo reflections more frequently at the surfaces/interfaces, so the 215 collisions with the surfaces/interfaces will become a significant fraction of the total 216 number of collisions. In addition, grain boundaries in polycrystalline wires may act 217 like partially reflecting planes for electron waves, so they also contribute to the 218 increase of resistivity. To investigate these effects directly, the conductivity 219 measurements by the four-tip STM at nanometer scales is very useful [29]. 220

Cu wires having the width between 70 nm and 1 µm were prepared using a Cu/Low-k 221 damascene processes which are now very common in semiconductor industry. Figure 6 222 shows the cross-sectional SEM images of the damascene structure made in SiO<sub>2</sub> layer. 223 Tantalum (Ta) was used as a barrier metal (BM) in this experiment. Cu damascene lines 224 were formed using conventional Cu process such as seed Cu, electrochemical plating 225 (ECP) Cu for trench filling, and chemical-mechanical polishing. The Cu nanowires are 226 not single-crystalline; they are consisted of small gains. By the electron back-scatter 227 diffraction (EBSD) method, such grains are visualized along the Cu damascene lines 228 [29]. The average grain size was measured to be about 100 nm at 70 nm wide Cu 229 nanowires, which did not change so much with the width, because the grain size is 230 thought to be determined by the height when the width is smaller than the height. 231

The four-tip STM was used to measure the resistance of individual wires as 232 shown in Fig. 7. By using Pt-coated CNT tips, the probe spacing can be reduced 233 down to the order of several ten nm routinely [23]. When the Pt-coated CNT tips 234 were used, the contact resistance between the tip and sample could not be smaller 235 than 50 k $\Omega$  because of its very small contact area. This means that it is impossible 236 to measure conductive materials whose resistance is less than 50 k $\Omega$  by the two-237 point probe method. Only with the four-point probe method, resistances much 238 smaller than the contact resistance (as small as  $0.1\Omega$  in the present case) can be 239 measured. Therefore, the combination of the CNT tips and the four-tip STM is 240 very powerful for studies in nanoscale measurements. 241





Fig. 7 a An SEM image of the Cu wire being contacted with the four probes [29]. b The enlargement of the area of rectangle in (a). c The illustration of the four-terminal I–V measurement

Four-terminal I–V measurements were performed by sweeping the bias voltage between the outer pair of tips and recording the current I and the voltage drop V between the inner pair of tips (Fig. 7c). The probe spacing between the voltage probes was reduced while measuring the I–V characteristics. The two current probes (tip 1 and tip 4) and one of the voltage probes (tip 2) were fixed during the measurements, and only tip 3 was moved between tip 2 and tip 4 (Fig. 7c).

The measured values of four-terminal resistance as a function of the probe 248 spacing between the contact points of the voltage probes on the Cu wires are 249 shown in Figs. 8 and 9. For all of them, the probe spacing dependence of resis-250 tance basically showed a linear one-dimensional feature, meaning a diffusive 251 transport. By multiplying the gradient of the fitted straight lines and the cross 252 section of Cu wires (which was estimated from SEM image in Fig. 6), the three-253 dimensional resistivity was calculated as 4.6, 3.7, and 3.4  $\Omega$  cm, for the 70, 50 nm, 254 and 1 µm wide Cu wires, respectively. The resistivity of Cu increased as the line 255 width decreased as shown in Fig. 9b. This result is understood by the Fuchs-256 Sondheimer theory for the surface-scattering effect and the Mayadas-Shatzkes 257 model for the grain boundary effect [29]. No change in the measured resistance 258

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Fig. 8 a The measured resistances of the 70 nm wide Cu wire are shown as a function of probe spacing. b The enlargement of the area under 600 nm of probe spacing [29]



Fig. 9 a The measured resistances of Cu wires are shown as a function of probe spacing for the 500 nm and 1  $\mu$ m wide wires. b Plot of the resistivity versus Cu line width [29]. The resistivity coming from the grain boundary scattering is assumed to be constant because the gran size is roughly independent of the line with in our samples

was found in each measurement before and after repeated contacts of CNT tips. Itmeans no significant damage on the sample by the probe contacts.

In this experiment, the probe spacing was reduced to the scale which is com-261 parable with the grain size. Therefore, it can be expected that there will be some 262 change in the resistance when the probe spacing becomes so short that electrons do 263 not undergo grain boundary scattering. This has been indeed observed. Figure 8b 264 shows the enlarged view of the data shown in Fig. 8a for the probe spacing smaller 265 than 600 nm. There is a slight jump in resistance when the probe spacing is shorter 266 than 200 nm. This directly corresponds to the grain boundary scattering where 267 additional resistance occurs at the grain boundary due to the reflection of electron 268 wave there. 269

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# 6 Concluding Remarks

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Electrical measurements using metal-coated CNT tips in the four-tip STM have been 271 demonstrated for the CoSi<sub>2</sub> nanowires and Cu damascene wires. Since the apex of 272 CNT tips is around 10 nm and the aspect ratio is so large, it is able to measure the 273 resistance at nanoscale surface areas. The minimum probe spacing in the four-point 274 probe resistance measurement was reduced to a few 10 nm, which is similar or less 275 than the grain size of polycrystalline wires and even the carrier mean free path. The 276 resistance along the wires present here increased linearly with the measured length, 277 meaning classical diffusive transport. But very recently, we have found quasi-278 ballistic transport in semiconducting FeSi2 nanowires at room temperature where the 279 carrier mean free path is much longer than that of the metallic wires. The details will 280 be reported elsewhere. In the case of Cu damascene wires, the resistivity increased as 281 the wire width decreased. This is due to the surface/interface scattering of carrier. We 282 have evaluated the surface/interface scattering quantitatively to obtain the specu-283 larity factor in Fuchs-Sondheimer theory. At the very narrow probe spacing which 284 was comparable to the grain size, the resistance jump due to a single grain boundary 285 was clearly observed. As demonstrated by the measurements presented here, the 286 four-tip STM with CNT tips is a very useful tool for transport physics at nanoscale as 287 well as industrial purposes. 288

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